

Examination of the structure of NaCl monocrystals with different orientations 5.4.13-00



What you can learn about ...

- Characteristic X-ray radiation
- Energy levels
- Crystal structures
- Reciprocal lattice
- Miller indices
- Bragg scattering
- Atomic form factor
- Structure factor

Principle:

Polychromatic X-rays are to be directed against NaCl monocrystals with different orientations. The spacing between the lattice planes of each monocrystals then to be determined by analyzing the wave-length-dependent intensity of the reflected radiation.

What you need:

X-ray basic unit, 35 kV	09058.99	1
Goniometer for X-ray unit, 35 kV	09058.10	1
Plug-in module with Cu X-ray tube	09058.50	1
Counter tube, type B	09005.00	1
Universal crystal holder	09058.02	1
Sodium chloride monocrystals, set of 3	09058.01	1

Recording equipment:

XYt recorder	11416.97	1
Connecting cable, $l = 100$ cm, red	07363.01	2
Connecting cable, $l = 100$ cm, blue	07363.04	2
or		
Software X-ray unit, 35 kV	14407.61	1
RS232 data cable	14602.00	1
PC, Windows® 95 or higher		

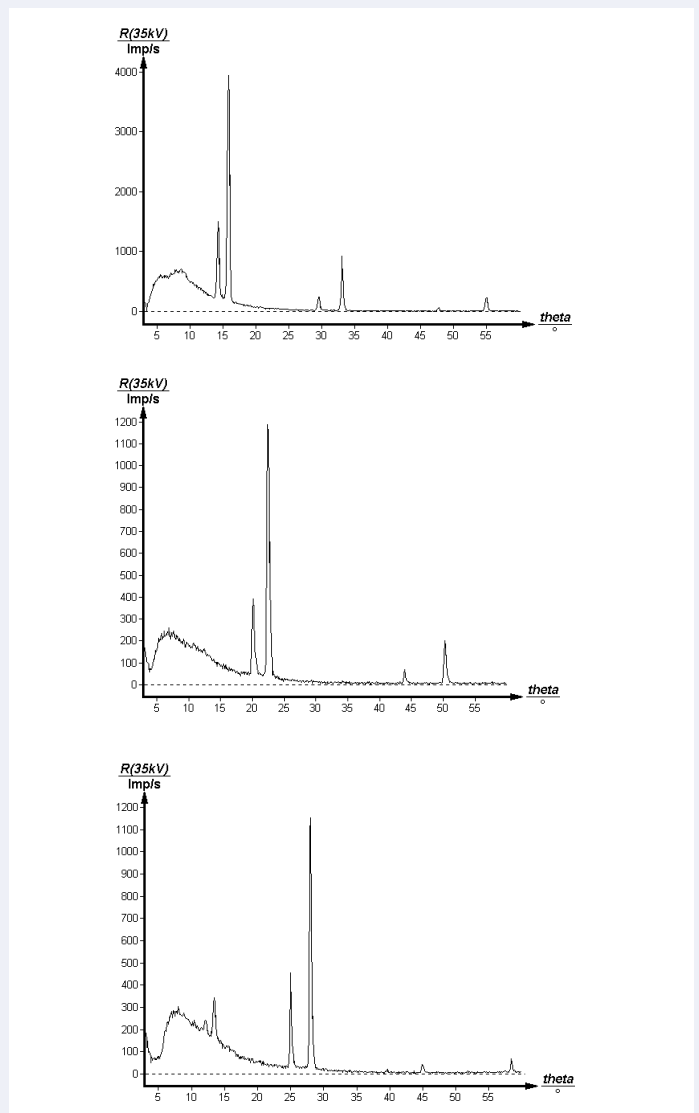
Complete Equipment Set, Manual on CD-ROM included
Examination of the structure of NaCl monocrystals
with different orientations
P2541300

Tasks:

1. NaCl monocrystals with the orientations (100), (110) and (111) are each to be separately used to record an intensity spectrum of the polychromatic radiation emanated by the X-ray tube.
2. The Bragg angles of the characteristic radiations are to be determined from the spectra, and the distances between lattice planes calculated for each orientation.
3. The planes of reflection and their Miller indices are to be found.

X-ray intensity of copper as a function of the glancing angle: NaCl monocrystal with different orientations as Bragg-analyzer:

1-(100); 2-(110); 3-(111)



Related topics

Characteristic X-ray radiation, energy levels, crystal structures, reciprocal lattice, Miller indices, Bragg scattering, atomic form factor, structure factor.

Principle

Polychromatic X-rays are to be directed against NaCl monocrystals with different orientations. The spacing between the lattice planes of each monocrystal then to be determined by analyzing the wavelength-dependent intensity of the reflected radiation.

Equipment

X-ray basic unit, 35 kV	09058.99	1
Goniometer for X-ray unit, 35 kV	09058.10	1
Plug-in module with Cu X-ray tube	09058.50	1
Counter tube, type B	09005.00	1
Universal crystal holder	09058.02	1
Sodium chloride monocrystals, set of 3	09058.01	1
 Recording equipment:		
XYt recorder	11416.97	1
Connecting cable, $l = 100$ cm, red	07363.01	2
Connecting cable, $l = 100$ cm, blue	07363.04	2
or		
Software X-ray unit, 35 kV	14407.61	1
RS232 data cable	14602.00	1
PC, Windows® 95 or higher		

Tasks

1. NaCl monocrystals with the orientations (100), (110) and (111) are each to be separately used to record an intensity spectrum of the polychromatic radiation emanated by the X-ray tube.
2. The Bragg angles of the characteristic radiations are to be determined from the spectra, and the distances between lattice planes calculated for each orientation.
3. The planes of reflection and their Miller indices are to be found.

Set-up and procedure

Set up the experiment as shown in Fig. 1. Fix the diaphragm tube with 2 mm diameter aperture in the X-ray outlet tube. With the X-ray unit switched off, connect the goniometer and the counter tube to the appropriate sockets in the base plate of the experimenting area. Set the goniometer block to the middle position and the counter tube to the right stop.

The following settings are recommended for the recording of the spectra:

- Auto and Coupling mode
- Gate time 2 s; Angle step width 0.1°
- Scanning range: 3° - 55°
- Anode voltage $U_A = 35$ kV; Anode current $I_A = 1$ mA

Fig. 1: Experimental set-up for the examination of the structure of NaCl monocrystals



Theory and evaluation

NaCl monocrystals have a face centred cubic lattice (fcc) (see Fig. 2). The basis of the cell is an Na⁺ ion at 000 and a Cl⁻ ion at $\frac{1}{2} \frac{1}{2} \frac{1}{2}$. For a cubic crystal with a lattice constant *a*, the following is valid for the distances *d*(*h,k,l*) of the individual lattice planes (*h,k,l*):

$$d = \frac{a}{\sqrt{h^2 + k^2 + l^2}} \quad (1)$$

When a beam of X-rays of wavelength λ impinges on a parallel layer of lattice planes (Fig. 3) at glancing angle ϑ , then, according to Bragg, constructive reflection occurs when condition (2) is fulfilled:

$$2d \sin \vartheta = n \lambda ; (n = 1,2,3,\dots) \quad (2)$$

The relative intensities of all reflections are determined by the scattering power and position of the individual atoms in the unit cell of the crystal, however, and are described by the structure factor *F*(*h,k,l*):

$$F(h,k,l) = \sum_n f_n \cdot \exp[-2\pi i(hu_n + kv_n + lw_n)] \quad (3)$$

where *f_n* = the atomic form factor, and *u_n*, *v_n* and *w_n* = the coordinates of the *n* atoms of a unit cell.

The total intensity of the scattered reflections is given by:

$$I = F^* \cdot F = |F(h,k,l)|^2 \quad (4)$$

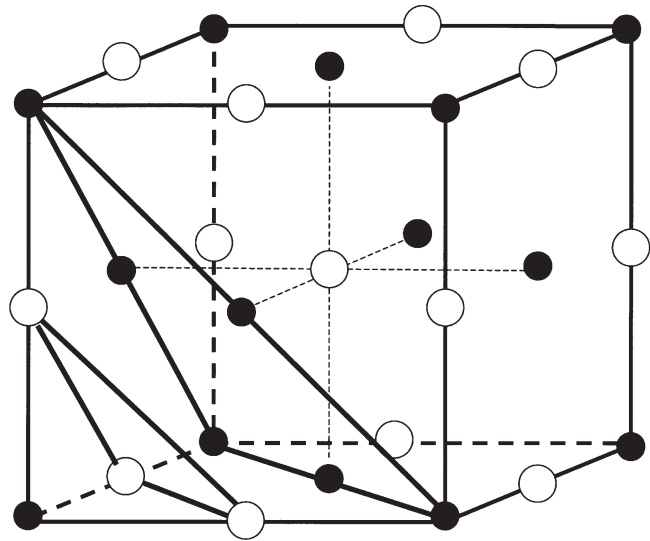


Fig. 2: NaCl crystal structure with drawn-in (111) and (222) lattice planes. Na⁺ ions = ● ; Cl⁻ ions = ○.

With the atom positions 000; $0\frac{1}{2} \frac{1}{2} \frac{1}{2}$; $\frac{1}{2} 0\frac{1}{2} \frac{1}{2}$; $\frac{1}{2} \frac{1}{2} 0$ as the basis of an fcc lattice, it follows from (3) that *F* = 0 when the *h,k,l* triplet contains even and odd numbers, and *F* = 4*f* when all indices are either even or odd.

Table 1

	(hkl)	$\vartheta(K_{\alpha})/^\circ$	$\vartheta(K_{\beta})/^\circ$	$d(\vartheta-K_{\alpha})/\text{pm}$	$d(\vartheta-K_{\beta})/\text{pm}$	<i>d</i> /pm	<i>d</i> (theor)/pm	Δ
(100) crystal								
<i>n</i> = 1	002	15.8	14.3	283.5	281.8	282.6 ₅		
<i>n</i> = 2	002	33.2	29.6	282.0	281.8	281.9		
<i>n</i> = 3	002	55.1	47.8	282.4	281.9	282.2 ₅		
						282.3	282.0	+0.1%
(110) crystal								
<i>n</i> = 1	022	22.5	20.1	201.7	202.5	202.1		
<i>n</i> = 2	022	50.3	43.9	200.6	200.6	200.6		
						201.3 ₅	199.4	+1%
(111) crystal								
<i>n</i> = 1	111	13.4	12.1	333.1	332.0	332.5 ₅		
<i>n</i> = 2	111	28.0	25.0	329.9	329.4	329.7 ₅		
<i>n</i> = 3	111	45.0	39.6	327.5	327.6	327.5 ₅		
<i>n</i> = 4	111	–	58.4	–	326.9	326.9		
						329.2	325.6	+1.1%

(Cu: $\lambda_{K\alpha} = 154.4 \text{ pm}$; $\lambda_{K\beta} = 139.2 \text{ pm}$)

The lattice constant for NaCl is $a = 564 \text{ nm}$. We therefore find the following distances between lattice planes:

$$d(200) = 282.0 \text{ pm}; d(220) = 199.4 \text{ pm}; d(111) = 325.6 \text{ pm}.$$

Figs. 4a-4c show the spectra of NaCl crystals with the orientations (100), (110) and (111).

Table 1 lists the Bragg angles calculated from the spectra, and the lattice planes calculated using (2).

The very good agreement between the value for the distance between the (100) lattice planes found practically and that calculated using (2) confirm the fact that ionic crystal can be very exactly split parallel to the (100) planes. The not quite so good agreement for crystals with (110) and (111) orientations is due to an inaccuracy of a few $1/10^\circ$ in the crystallographic directions. It can be further seen that an evaluation of the reflections leading to agreement with the measured values is only possible when the assumption is made, that only lattice planes with either even or odd Miller indices interfere constructively.

The spectrum of the NaCl crystal with (111) orientation is worthy of attention. The intensity of the first order (111) reflection is unusual in that it is distinctly lower than that in the second order. This is because, in this orientation, the lattice planes are alternately occupied by either only Cl^- ions or only with Na^+ ions (Fig. 2). As these two ions have different scattering powers (atomic form factors), the intensities of their reflections differ.

Fig. 4: X-ray intensity of copper as a function of the glancing angle: NaCl monocrystal with different orientations as Bragg-analyzer: 4a-(100); 4b-(110); 4c-(111)

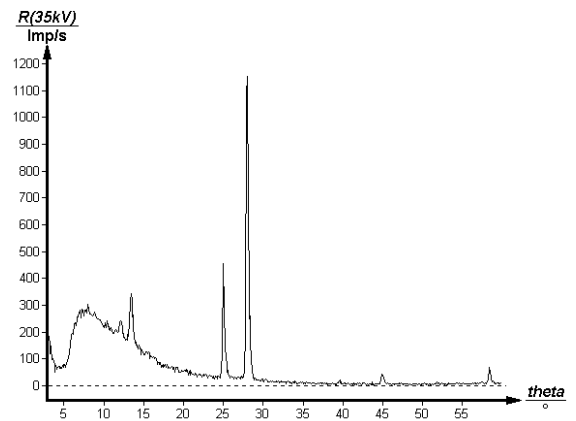
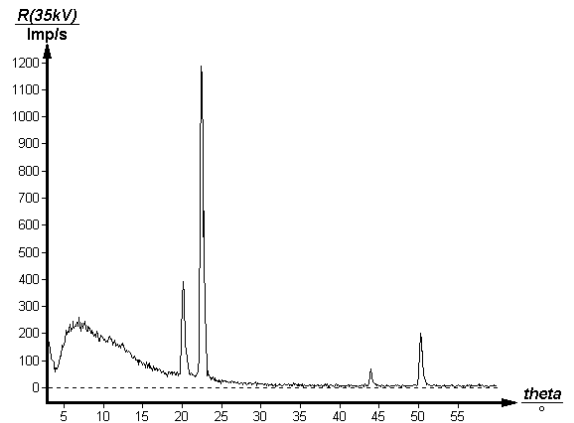
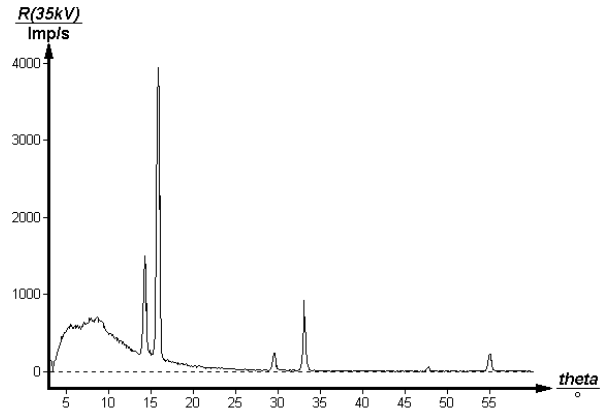
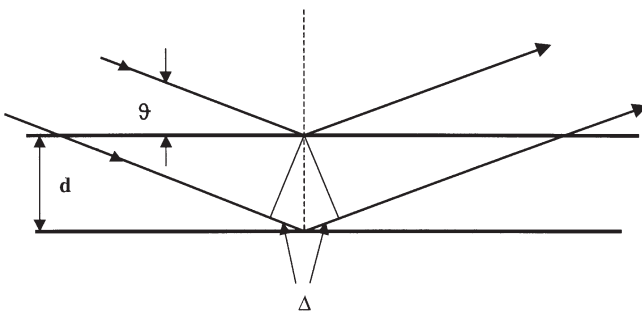


Fig. 3: Bragg scattering on the lattice planes



LEP
5.4.13
-00

**Examination of the structure of NaCl monocrystals
with different orientations**

